

Thermal, Conductivity, Structural and Optical Properties of $\text{HgI}_2\text{-As}_2\text{X}_3$ ($\text{X} = \text{Se}, \text{Te}$) Chalcogenide Glasses

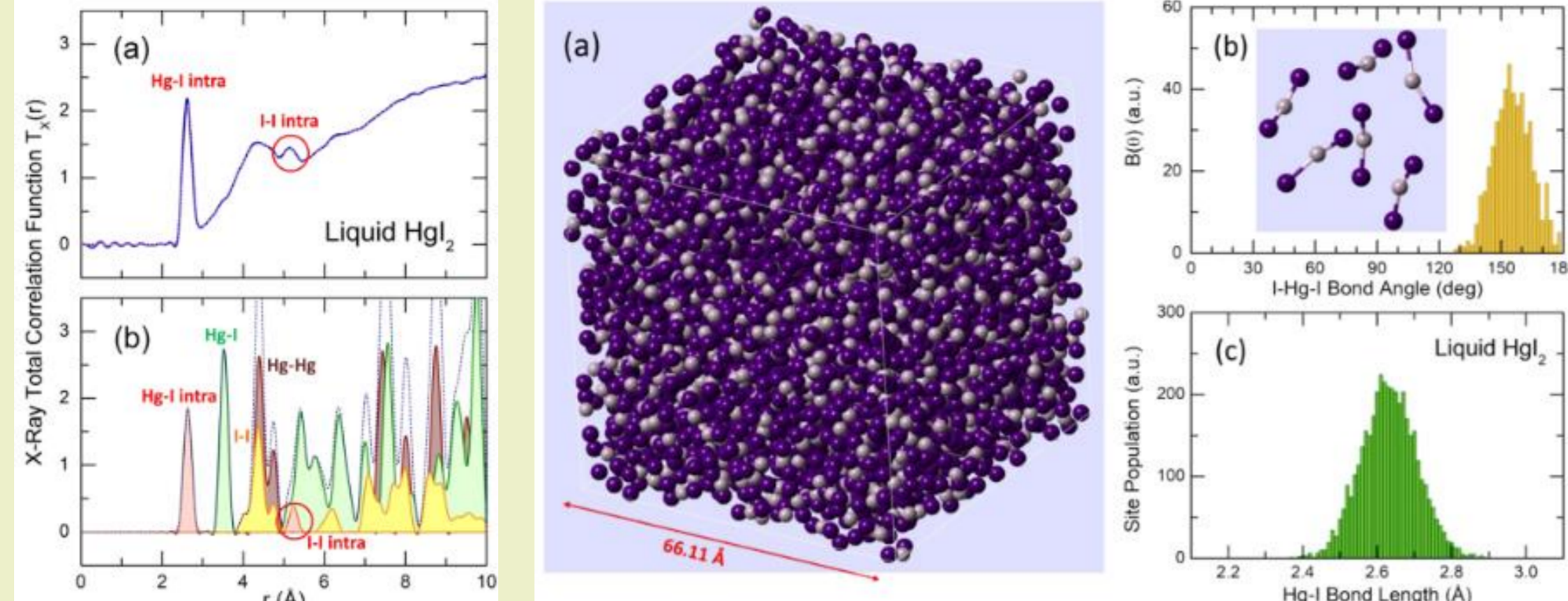
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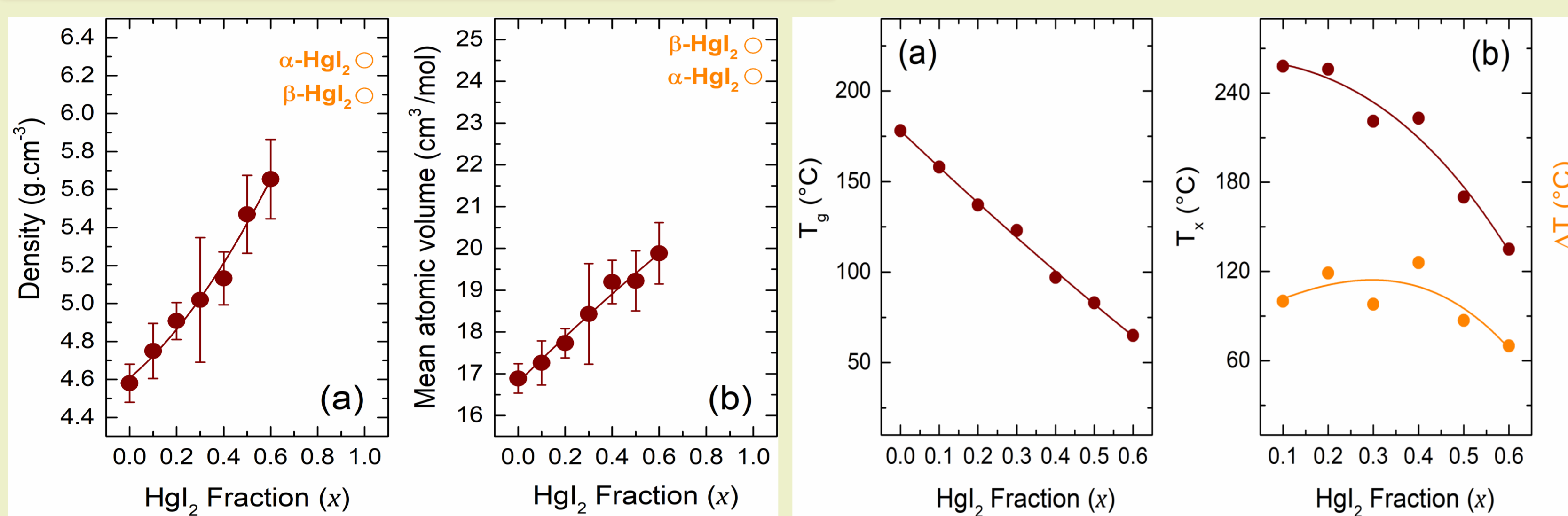
I. Introduction

Crystalline mercury iodide HgI_2 exists in several polymorphs: (1) Layered red $\alpha\text{-HgI}_2$ and metastable orange polymorphs build-up of centrosymmetric CS-HgI_4 tetrahedra, (2) Orthorhombic yellow $\beta\text{-HgI}_2$, existing above the $\alpha \rightarrow \beta$ phase transition at 135 °C, composed of centrosymmetric quasi-linear HgI_2 triatomic molecules; $\angle\text{I-Hg-I} = 178.3^\circ$, and (3) yellow^{HT} form consisting of non-centrosymmetric bent

HgI_2 molecules; $\angle\text{I-Hg-I} = 160^\circ$. The question arises whether (1) mercury will modify the trigonal networks of the As_2X_3 ($\text{X} = \text{Se}, \text{Te}$) chalcogenide glasses and (2) HgI_2 entities are non-centrosymmetric.

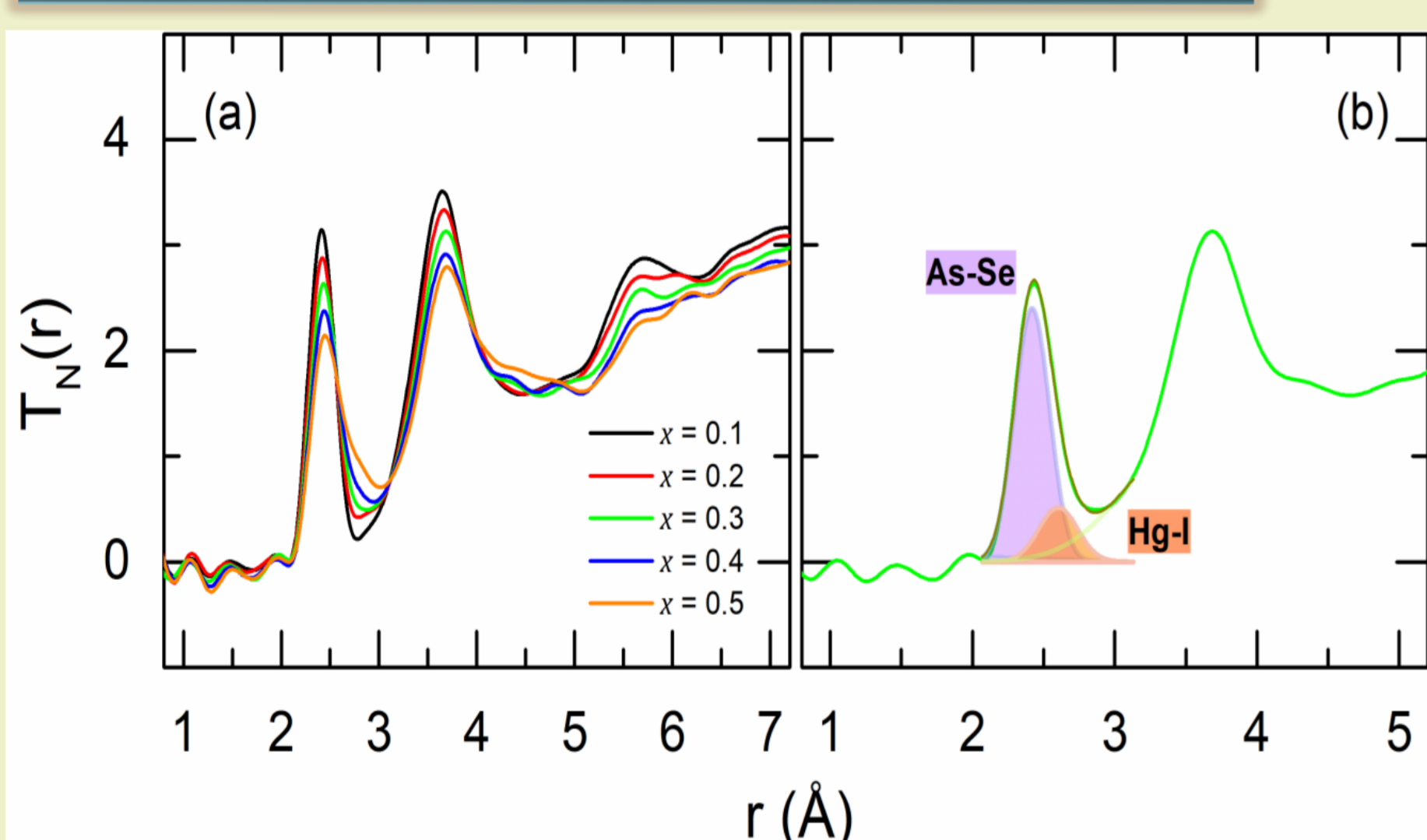


II. Macroscopic properties



- ✓ The density & Mean atomic volume increase monotonically with increasing HgI_2
- ✓ The reduced network connectivity in the $\text{HgI}_2\text{-As}_2\text{Se}_3$ glasses seems to be responsible for T_g decrease

V. Neutron diffraction



Hypothesis:

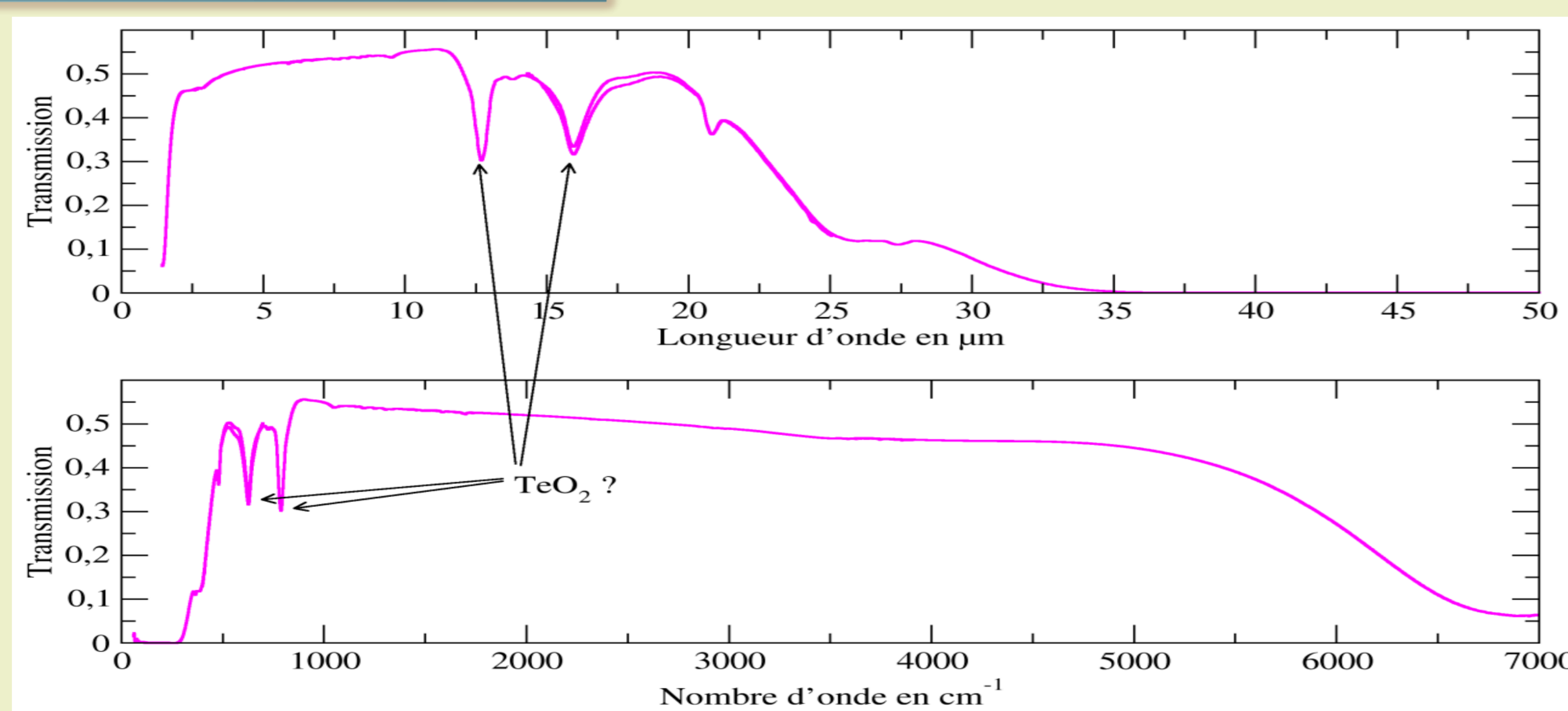
□ two or more different mercury iodide local environments:

- 1) Hg-I molecular chains
- 2) tetrahedral $\text{HgI}_{4/4}$ -related motifs
- 3) a mixture of both

HgI_2 Fraction x	$r(\text{As-Se})$ (Å)	$N_{\text{As-Se}}$	$r(\text{Hg-I})$ (Å)	$N_{\text{Hg-I}}$
0.0	2.41(1)	2.95(10)	-	-
0.1	2.41(1)	3.05(10)	2.58(2)	2.85(10)
0.2	2.41(1)	3.03(10)	2.57(2)	2.62(10)
0.3	2.41(1)	3.01(10)	2.60(2)	2.82(10)
0.4	2.42(1)	2.95(10)	2.64(1)	3.00(10)
0.5	2.42(1)	3.00(10)	2.67(1)	3.36(10)

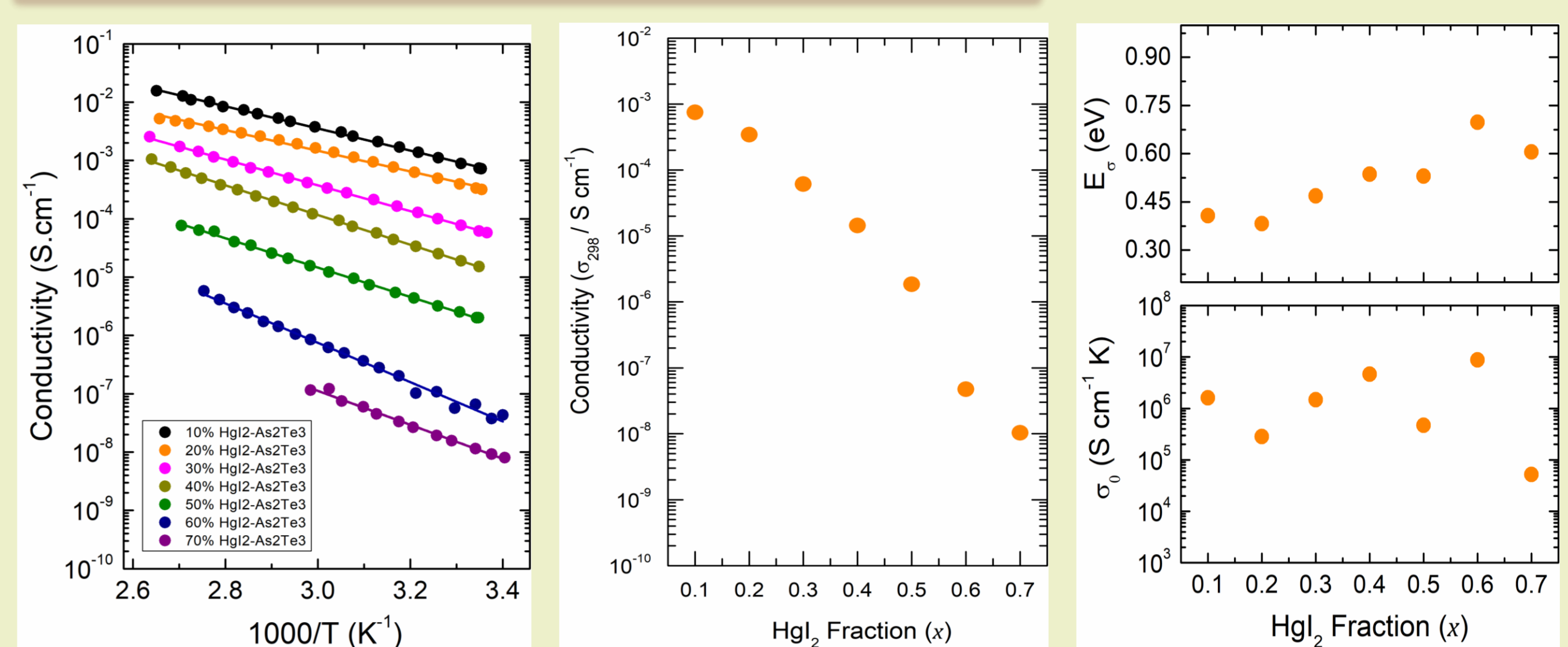
VI. Optical properties

$\text{HgI}_2\text{-As}_2\text{Te}_3$:



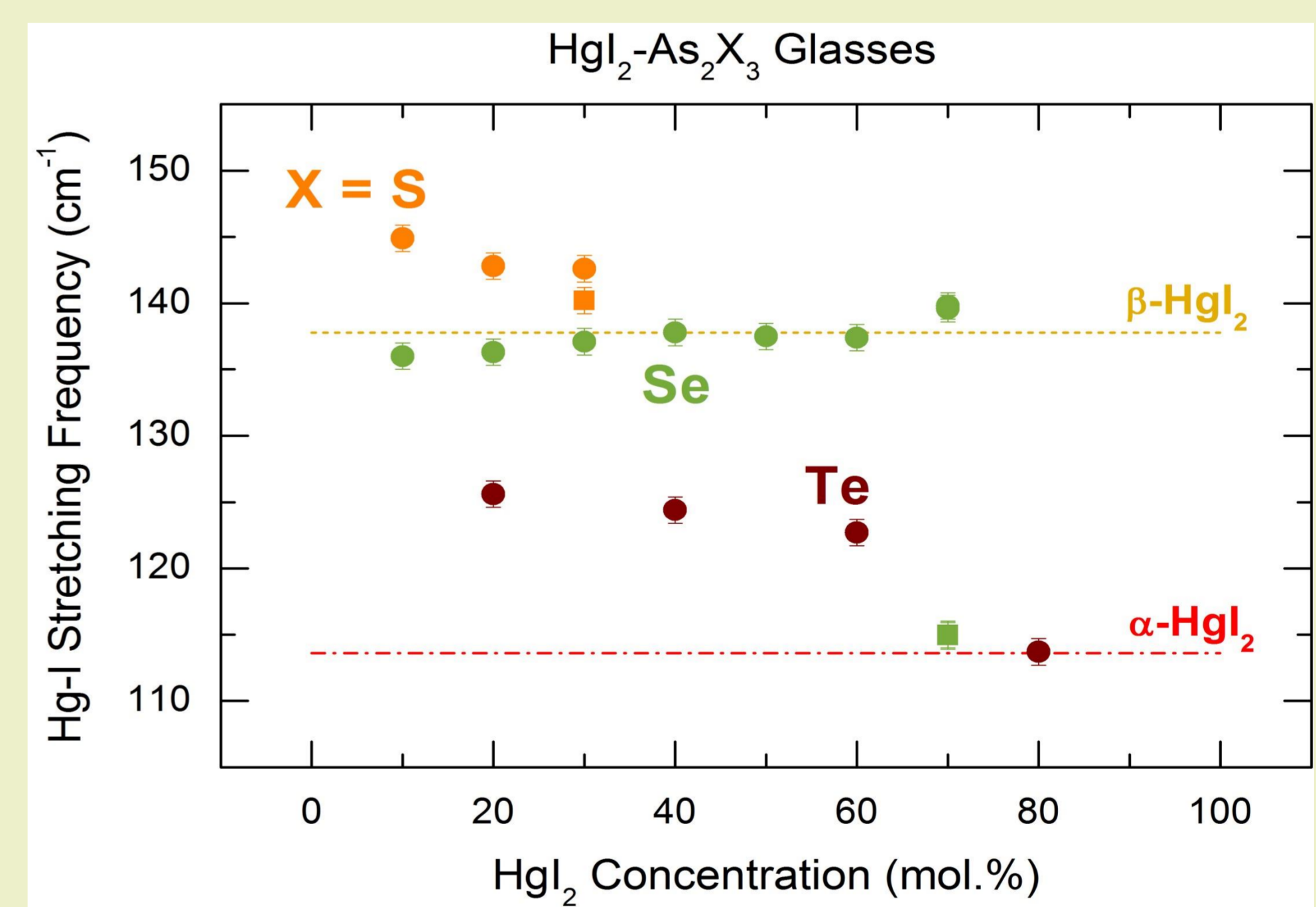
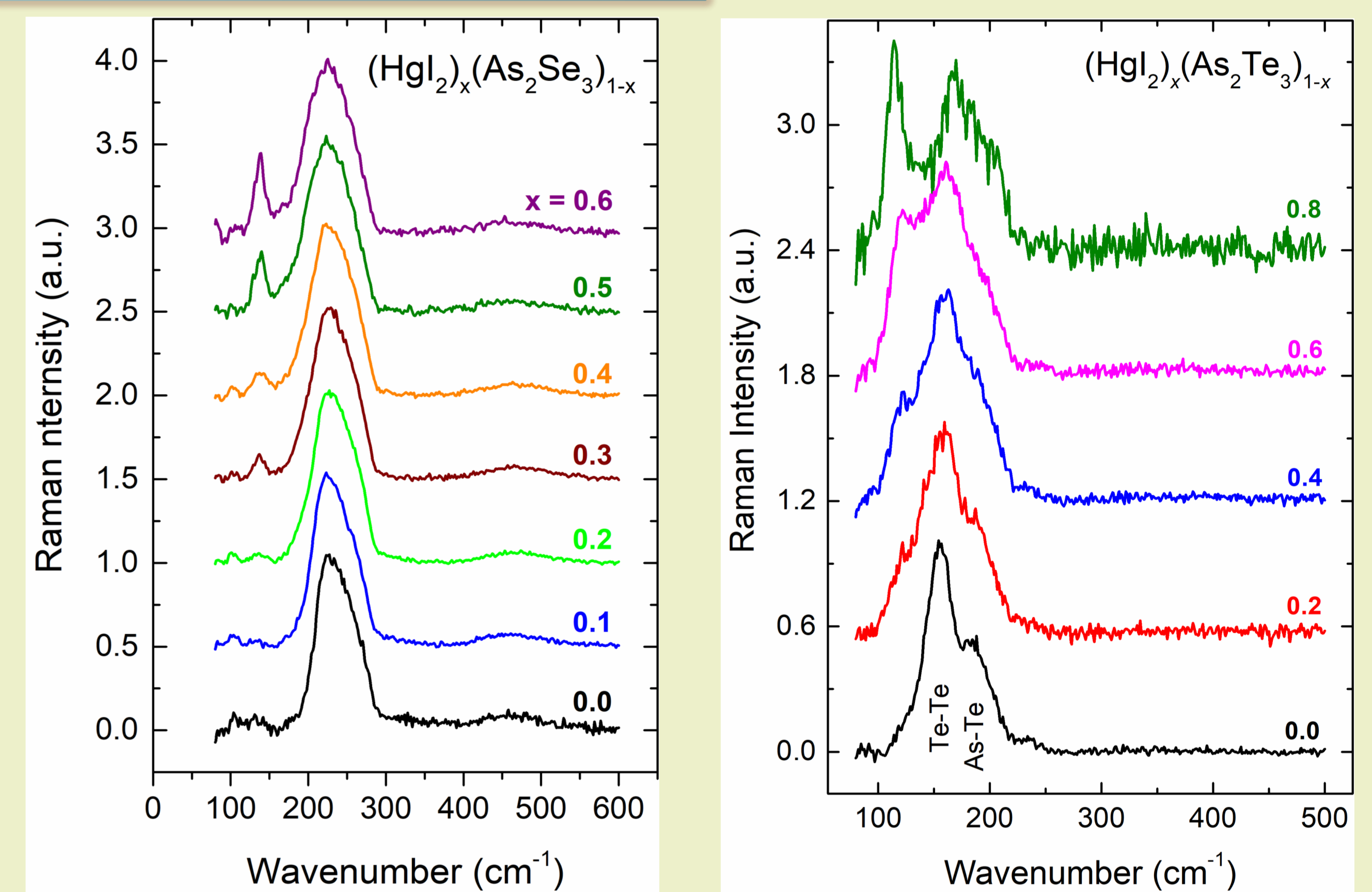
✓ long infrared spectral range up to 35 μm

III. Conduction properties



- ✓ The conductivity obeys the Arrhenius law: $\sigma = \sigma_0 / T \exp(-E_a/kT)$
- ✓ Highly energetic Te 5p lone pairs decreasing the gap is responsible for high electronic conductivity in telluride glasses
- ✓ Decreasing tellurium concentration with increasing x results in a gap widening and thus room-temperature $\sigma_{298}(x)$ decrease
- ✓ Overall increase of activation energy with increasing x

IV. RAMAN Studies



$\text{HgI}_2\text{-As}_2\text{Se}_3$:

- ✓ 227 cm^{-1} \Rightarrow stretching modes in $\text{AsSe}_{3/2}$ trigonal units
- ✓ 270 cm^{-1} \Rightarrow Se - Se stretching modes of As-Se-Se-As units
- ✓ 139 cm^{-1} \Rightarrow similar to A_{1g} symmetric Hg-I stretching in $\beta\text{-HgI}_2$

$\text{HgI}_2\text{-As}_2\text{Te}_3$:

- ✓ 155 cm^{-1} \Rightarrow Te - Te related chains/fragments in the As-Te glass
- ✓ 190 cm^{-1} \Rightarrow stretching modes in $\text{AsTe}_{3/2}$ trigonal units
- ✓ 114 cm^{-1} \Rightarrow Hg-I stretching in $\alpha\text{-HgI}_2$ / 123 cm^{-1} ($0.0 \leq x \leq 0.6$)

VII. Perspective

- ✓ DFT modelling of the Raman spectra
- ✓ High-energy X-ray diffraction (HE-XRD) for the $\text{HgI}_2\text{-As}_2\text{S}_3$ glasses
- ✓ Structural studies of the $\text{HgI}_2\text{-As}_2\text{Te}_3$ glasses using neutron diffraction (ND) and HE-XRD
- ✓ First-Principle Molecular Dynamics (FPMD) to decipher the structural motifs within the glass network
- ✓ Infrared optical measurements